

Spectroscopic and Electrical properties of Polyaniline/ Y_2O_3 composites and their application as Humidity sensor

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Abstract—Conducting Polyaniline/Yttrium oxide (PANI/ Y_2O_3) composites have been synthesized by insitu deposition technique by placing fine graded Y_2O_3 in polymerization mixture of aniline. The results were also well supported by X-ray diffractometry (XRD), Infrared (IR) spectral analysis, scanning electron microscope (SEM) and conductivity measurements. It is observed from the XRD studies that Y_2O_3 has retained its structure even though it is dispersed in PANI during polymerization reaction. By observation of IR spectra, it is seen that some of the characteristic stretching frequencies are considerably shifted towards higher frequency side. The surface morphology of these composites was studied by SEM and it was found that, composites possess grains and porous structure. High conductivity measurements show thermal activated behavior. The change in resistance with respect to percentage relative humidity (% RH) is observed. The composites in the pellet form exhibit almost linear behavior within a chosen range of humidity (ranging between 10 and 95% RH)..

Index Terms— Polyaniline composites; Y_2O_3 ; conductivity; Humidity sensors

I. INTRODUCTION

For the past two decades, conducting polymers have enjoyed significant research interest, both pure and applied [1-5]. The latter has its stimulation from the application potential of conducting polymers in solid state devices such as solar cells, Schottky junctions, displays, sensors, and microelectronics. However, many conducting polymers have poor mechanical properties and environmental stability which make them unsuitable for device fabrication. To overcome this difficulty, efforts have been made to synthesize various composites in which conducting polymers are embedded in an insulating polymer matrix having good mechanical properties [6-9]. The study of electrical transport in conducting polymers has been an interesting area of research. During the injection of charge carriers with various dopants like ClO_4 , BF_4 , etc. the polymer chain gets easily distorted around the injected charge and polaron is formed. At low doping levels polarons are the only charge carriers and as doping level increases, two polarons come close to each other resulting in the formation of bipolarons [10].

For using a composites material incorporating conducting polymers in device fabrication, it is necessary to know the nature of charge carriers, conduction mechanism, and the effect of other phases present in the system. Frequency dependent measurement is a useful tool

to understand the nature of charge carriers (polarons and bipolarons) and the conduction mechanism [11]. Epstein and co-workers [12] have carried out such measurements on poly acetylene. They have suggested that charge transport occurs by charged solitons. In particular, at low doping level, an intersoliton electron hopping mechanism has been proposed by Kivelson [13].

Among all conducting polymers polyaniline (PANI) achieved widespread importance because of its unique conduction mechanism and environment stability. The survey of literature reveals that the detailed conductivity studies on PANI/ Y_2O_3 are scarce. In the present study, PANI and polyaniline yttrium oxide (PANI/ Y_2O_3) composite (with varying weight percentage of yttrium oxide in polyaniline) have been synthesized. These samples are characterized by the various techniques such as XRD, FTIR, SEM and electrical properties are measured and studied.

2. Experimental

2.1 Synthesis of PANI/ Y_2O_3 composites

Aniline (AR grade) was purified by distillation before use and ammonium per sulphate $[(NH_4)_2S_2O_8]$, HCl were used as received. 0.1 mol aniline monomer is dissolved in 1 mole hydrochloric acid to form polyaniline. Fine graded pre-sintered yttrium oxide (AR grade, SD-Fine

Chem.) powder in the weight percentages (wt %) of 10, 20, 30, 40 and 50 is added to the polymerization mixture with vigorous stirring in order to keep the yttrium oxide powder suspended in the solution. To this reaction mixture, $[(\text{NH}_4)_2\text{S}_2\text{O}_8]$ which is used as oxidant is added slowly drop-wise with continuous vigorous stirring for the period of 4-6 hours at temperature $0-5^\circ\text{C}$. Polymerization of aniline takes place over fine grade yttrium oxide particles. The resulting precipitate is filtered under suction and washed with distilled water until the filtrate becomes colorless. Acetone is used to dissolve any unreacted aniline. After washing, the precipitate is dried under dynamic vacuum at $60-80^\circ\text{C}$ for 24 hrs to get resulting composites. In this way five different polyaniline yttrium oxide composites with different weight percentage of yttrium oxide (10, 20, 30, 40 and 50) in polyaniline have been synthesized. All the composites are crushed into fine powder in an agate mortar in the presence of acetone medium. The composite powder is pressed to form pellets of 10mm diameter and thickness which varies from 2 to 2.75 mm. The electrical measurements on these samples were made using the silver paint as electrodes on both sides.

2.2 Measurements.

X-ray diffraction patterns of PANI/ Y_2O_3 were obtained with Philips X-ray diffractometer using $\text{CuK}\alpha$ ($\lambda = 1.5404\text{\AA}$). The Fourier transform infrared (FTIR) spectra of the composites were recorded on Perkin Elmer (model 783) IR spectrometer in KBr medium at room temperature, in the region $4000 - 450\text{ cm}^{-1}$. The powder morphology of PANI and PANI/ Y_2O_3 samples in the form of pellets was investigated using Philips XL 30 ESEM scanning electron microscope (SEM). The dc conductivity of the PANI/ Y_2O_3 composites were measured by two probe technique at temperature range 40 to 150°C . The planar resistance of the sensors was recorded by controlling the humidity in a closed chamber at room temperature.

3. Results and discussion.

3.1 X-ray diffraction (XRD)

Figure 1 shows XRD patterns for PANI/ Y_2O_3 composite with 50 wt% of Y_2O_3 in PANI. It is seen from figure 1, that the cubic peak of Y_2O_3 indicates the crystalline nature of the composite. By comparing the XRD pattern of the composite with that of Y_2O_3 , the prominent peaks corresponding to $2\theta = 29.79, 34.52, 49.19$ and 58.25 are due to $(2\ 2\ 2)$, $(4\ 1\ 1)$, $(4\ 4\ 0)$ and $(6\ 2\ 2)$ planes of Y_2O_3 [JCPDS file no. 41-1105]. By comparing the XRD pattern of the composite and Y_2O_3 , it is confirmed that Y_2O_3 has retained its structure even

though it is dispersed in PANI during polymerization reaction.

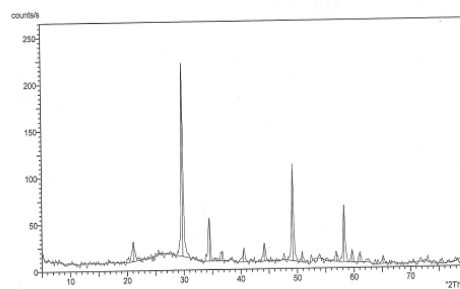


Figure 1. XRD patterns for PANI/ Y_2O_3 composite with 50 wt% of Y_2O_3 in PANI.

3.2 FTIR Spectra

Figure 2 shows the FTIR spectra of PANI/ Y_2O_3 composite (50% wt of Y_2O_3 in PANI). The FTIR transmission spectra of powder using KBr pellets having different weight percentage of Y_2O_3 in PANI were recorded in the range $450 - 4000\text{ cm}^{-1}$ to confirm polymerization of polyaniline. The spectra for all the samples showed strong bands in the region $470 - 1800\text{ cm}^{-1}$ that are the characteristics of PANI [14]. Similar

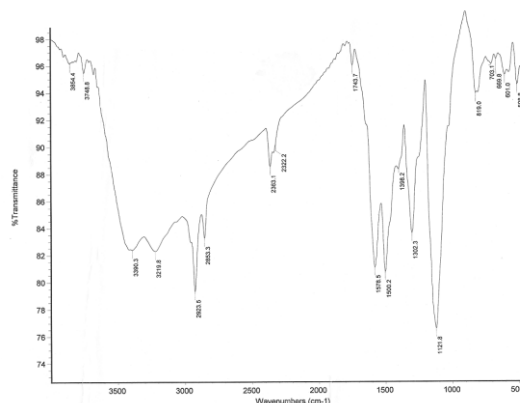


Figure 2. FTIR spectra of PANI/ Y_2O_3 composite (50% wt of Y_2O_3 in PANI).

stretching frequencies can also be found in other composites but intensity of metal oxygen peak increases as the weight present of Y_2O_3 is increased. The important peaks that are observed in this composite in FTIR spectra are observed at $1743\text{ cm}^{-1}, 1578\text{ cm}^{-1}, 1500\text{ cm}^{-1}, 1398\text{ cm}^{-1}, 1302\text{ cm}^{-1}, 1121\text{ cm}^{-1}, 819\text{ cm}^{-1}, 703\text{ cm}^{-1}, 668\text{ cm}^{-1}, 601\text{ cm}^{-1}, 508\text{ cm}^{-1}$ and 470 cm^{-1} . By observation of IR spectra, some of the characteristic stretching frequencies are considerably shifted towards higher frequency side. The typical peaks observed are at $1578\text{ cm}^{-1}, 1302\text{ cm}^{-1}, 1121\text{ cm}^{-1}, 819\text{ cm}^{-1}$ and 508 cm^{-1} which may be attributed due to the Vander walls kind of interaction between Y_2O_3 and PANI chain [15].

3.3 Scanning electron Micrograph (SEM).

SEM of PANI/ Y_2O_3 composites with 50 wt% of Y_2O_3 in PANI is shown in figure 3. High magnification SEM image reveals the presence of Y_2O_3 particles uniformly distributed throughout the composite sample. The composite possess grain and porous structure, and, further the composites have capillary pores connected by pores. The SEM studies performed on all the samples indicated a transformation from a cluster pattern to branched chain structure (or a fibrillar morphology) with increase in wt% of yttrium oxide [16]. The contrast in this image is due to the difference in scattering from different surface areas as a result of geometrical difference between PANI and Y_2O_3 .

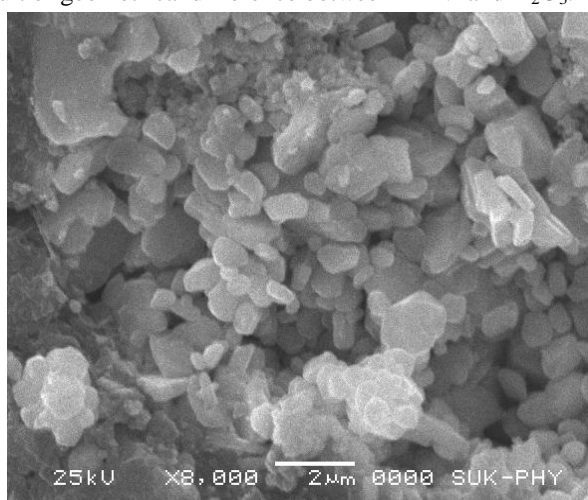


Figure 3. SEM of PANI/ Y_2O_3 composites with 50 wt% of Y_2O_3 in PANI

3.4 Temperature dependent conductivity.

Figure 4 shows the temperature dependent conductivity of PANI/ (50 wt% of Y_2O_3 in PANI) composites. From the figure, it is observed that the conductivity is found to be increase with temperature. Among the five composites, PANI/ Y_2O_3 (50 wt% of Y_2O_3 in PANI) was seen to yield with highest conductivities. The increase in conductivity with increase in temperature is the characteristic of 'thermal activated behavior'. The increase in conductivity would be due to the increase of efficiency of charge transfer between Y_2O_3 and polymer chains with increase in temperature [15, 16]. It is also be suggested that the thermal curing effects the chain alignment of the polymer, which leads to the increase of conjugation length and that brings about the increase of conductivity. Also, there had to be molecular rearrangement on heating, which made the molecular conformation favorable for electron delocalization [17].

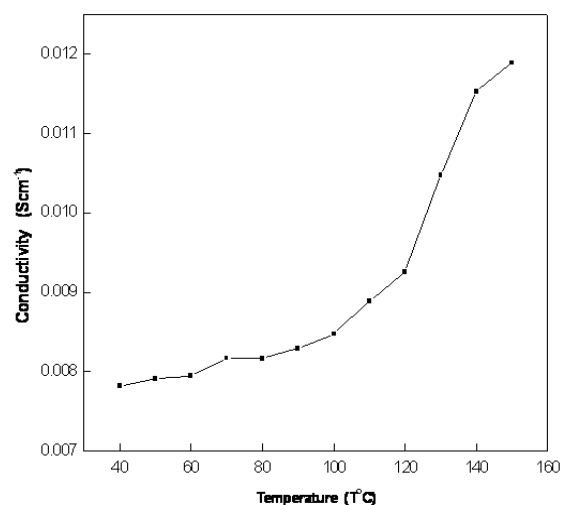


Figure 4. Temperature dependent conductivity of PANI/ (50 wt% of Y_2O_3 in PANI) composites.

3.5 Humidity Sensing Property.

Figure 5 shows the characteristic response of the PANI/ Y_2O_3 composites (10, 30 and 50 wt% of Y_2O_3 in PANI) as a function of relative humidity (% RH). From the figure, it can be noted that the resistance is varying almost linearly from 20 to 90% RH and is found to decrease from low humidity (dry state) to high humidity (wet state). At low humidity, when adsorption starts on the clean oxide surface, a layer of hydroxyl group is formed. The water vapor molecules are chemisorbed through a dissociative mechanism by which two surface hydroxyls per water molecule are formed. At higher relative humidity, the water molecule adsorbed on the PANI/ Y_2O_3 composite (50 wt% of Y_2O_3 in PANI) having grains and capillary pores enhance the electrolytic conduction as well as the protonic conduction by permitting capillary condensation of water molecules within the pores. The response time for sensing the humidity for an increase in relative humidity values by 10% was measured to be about 4-5s, wherein it recovers back to the normal value in around 10 s. The samples were found to be stable for more than 1 year.

The decrease in resistance or increase in the conductivity with increasing humidity can be attributed to mobility of the Y_2O_3 ion, which is loosely attracted to the polymer chain by weak Vander-walls force of attraction. At low humidity, the mobility of the Y_2O_3 ion is restricted because under dry conditions the polymer chains would tend to curl up into compact coil form. On the contrary, at high humidity, the polymer absorbs water molecules and gets hydrated, followed by the uncurling of the compact coil form into straight chains that are aligned with respect to each other. This geometry of the polymer is favorable

for enhanced mobility of the Y_2O_3 ions or the charge transfer across the polymer chains and hence the conductivity.

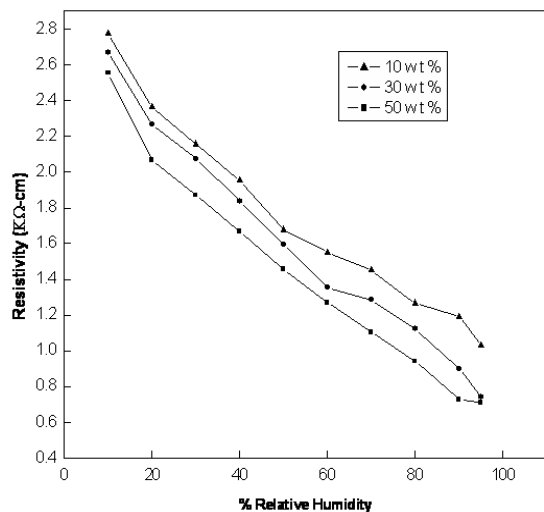


Figure 5. Characteristic response of the PANI/ Y_2O_3 composites (10, 30 and 50 wt% of Y_2O_3 in PANI) as a function of relative humidity (% RH).

Also, it has been reported that the conductivity of a conducting polymer increases when the sample absorbs the moisture. Decrease of resistance with the increase in the humidity proves the adsorption of the water molecules, which makes the polymer more p-type in nature, that is, the whole concentration is increased by the donation of lone pair from the conducting complex towards the Y_2O_3 water molecules. Thus, the partial charge transfer process of conducting species with that of water molecules results in the decrease of sheet resistivity. At higher humidity level, the mechanism may be different. The almost linear variation with respect to percentage relative humidity (%RH) can be used in an amplifier circuit for converting the measured values into measurable % RH values. On careful observation of figure 5, it is clearly seen that the PANI/ Y_2O_3 (50 wt% of Y_2O_3 in PANI) composite shows a linear response from 20 to 90% RH. On the other hand, in PANI/ Y_2O_3 composites (10 and 30 wt% of Y_2O_3 in PANI), the resistance is found to drop down from 20 up to 50% RH in a linear fashion while, after 60% RH slightly saturation is observed with a very small decrease in the resistance up to 95% RH. In other words, we can say that PANI/ Y_2O_3 (10 and 30 wt% of Y_2O_3 in PANI) shows two step sensing response. Thus PANI/ Y_2O_3 (50 wt% of Y_2O_3 in PANI) shows better sensing properties and exhibits good linearity in sensing response curve [18]

4. Conclusion

In this paper, we have presented a humidity sensor based on polyaniline/Yttrium oxide (PANI/ Y_2O_3) composites synthesized by 'insitu' polymerization. This is the novel polymerization process for the direct synthesis of emeraldine phase of the polymer. Formation of mixed phases of polymer together with conducting emeraldine salt phase is confirmed for the spectroscopic techniques. High temperature conductivity measurements show thermal activated behavior. The almost linear response of PANI/ Y_2O_3 (50wt% of Y_2O_3 in PANI) to the broad range of humidity proves to a competent material as humidity sensor.

REFERENCES

- [1] Hand book of Conducting polymers, edited by T.A.Skotheim(Marcel Dekker, NewYork, 1986).
- [2] S.Stafsform, J. L. Bredas, A. J. Epstein, H.S Woo, D. B. Tanner, W S Huang, and A. G.MecDiarmid, Phys. Rev.Lett.59,1446 (1987).
- [3] A. J. Heeger, S. Kivelson, J. R. Schrietter, and W. P. Su, Rev. Mod. Phys. 60, 781 (1988).
- [4] Y. Cao, P. Smith, and A. J. Heeger, Synth. Met. 32, 263 (1989).
- [5] A. G. MacDiarmid and A. J. Eppstein, J. Chem. Soc., Faraday Trans. 5, 1 (1989).
- [6] M. A. De Paoli, R. J. Waltman, A. F. Diaz, and J. Bargon, J. Chem. Soc. Chem. Commun., J. Polym. Sci. Polym Chem. 2d. 23,1687 (1985).
- [7] O.Niwa and T. Tamamura, J. Chem. Soc. Chem. Commun., 817 (1984).
- [8] V. Bocchi and G. P. Gardini, J. Chem. Soc. Chem. Commun., 148 (1986).
- [9] J. Ronacali and F. Garnier, J. Phys.: Condens. Matter 92, 833 (1988).
- [10] J. L. Bredas, J. C. Scott, K. Yakushi, and G. B. Street, Phys. Rev. B 30, 1023 (1984).
- [11] S. R. Elliott, Adv. Phys. 36, 135,(1987).
- [12] A. J. Epstein, H. W Gibson, P. M. Chaikin, W. G. Clark, and Grilnev, Phys. Rev. Lett. 45, 1730 (1980).
- [13] S.Kivelson, Phys. Rev.Lett 46,1344(1981).
- [14] Somani P, Marimuthu R, A.B. Mandale, Synthesis, Characterization and transport mechanism in conducting polyaniline / V_2O_3 composites , Polymer 42, 2991-3001 (2001).
- [15] R H Bughmass and L W Shacklette Phys. Rev.B29, 5872 (1989).
- [16] Suri, K.; K Annapoorni, S.; Sarkar, A. K.; Tandon, R. P. Sens Actuators B 81,227 (2002).
- [17] N.F. Mott, E.A Davis, Electron processes in Non-linear Materials.
- [18] M. Raghu, S.V. Subramanyam, Phys. Rev. B 43, 4236 (1991).